



# High-yield electrochemical upgrading of CO<sub>2</sub> into CH<sub>4</sub> using large-area protonic ceramic electrolysis cells

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## ABSTRACT

Electrochemical production of commodity chemicals via CO<sub>2</sub>-H<sub>2</sub>O co-electrolysis using solid oxide electrolysis cells presents a promising cost-effective energy-storage approach. Here, we harness the unique property of protonic ceramic electrolysis cells (PCEC) and demonstrate direct electrochemical production of CH<sub>4</sub> from CO<sub>2</sub>-H<sub>2</sub>O in a PCEC unit-cell stack. An exceptional CH<sub>4</sub>-yield ratio of 34.6% from only CO<sub>2</sub>-H<sub>2</sub>O reactants and greater than 70% with exhaust H<sub>2</sub> recycle were achieved under an electrolysis current of  $-1 \text{ A cm}^{-2}$  at 450 °C. Additionally, the electrochemical co-conversion of CO<sub>2</sub>-H<sub>2</sub>O offered a higher CH<sub>4</sub>-yield ratio compared to the thermochemical conversion of CO<sub>2</sub>-H<sub>2</sub> under certain operating conditions, indicating possible electrochemical promotion of catalytic CO<sub>2</sub> methanation. Techno-economic analyses were conducted to reveal potential operating conditions that yield a promising leveled cost of fuel production. The demonstrated good performance of the unit-cell stack shows promising scalability of PCECs for practical application from a system-level viewpoint.

## 1. Introduction

The market emergence of site-specific and intermittent renewable energy sources, such as wind and solar energy, necessitates the development of energy-storage technologies. Energy storage in the form of economically fungible chemical fuels potentially provides a low-cost grid-balancing solution for renewables penetration. Solid oxide electrolysis cells (SOECs) can efficiently electrochemically upgrade water and/or carbon dioxide and produce commodity chemicals, such as hydrogen, syngas or hydrocarbon fuels, for future on-demand power generation [1,2]. The chemical fuels can be either compressed and stored in tanks, or directly transported through pipelines, enabling large storage capacity and nearly infinite energy-storage duration. Here we harness the unique properties of protonic ceramic electrolysis cells (PCECs) for electrochemical CO<sub>2</sub> upgrading to a commodity chemical, CH<sub>4</sub>, that can be easily transported to end-users for subsequent on-demand power generation. This energy-storage solution brings the added benefit of CO<sub>2</sub> reuse and the potential alleviation of greenhouse

gas emissions.

Although energy storage with H<sub>2</sub> as the energy carrier has been widely studied, the use of CH<sub>4</sub> as the energy carrier brings thermodynamic and performance advantages [3–5]. Major hurdles facing hydrogen include its low volumetric energy density, the need for safe and cost-effective storage and transportation solutions, and its limited compatibility with existing infrastructure, such as natural gas pipelines and end-use appliances [6]. Methane handling is more straightforward; the CH<sub>4</sub>-based product gas can be directly introduced into the existing natural gas infrastructure. Most importantly, using CH<sub>4</sub> as an energy carrier enables a higher cell-level thermodynamic maximum round-trip efficiency ( $\epsilon_{\text{RT,max}}$ ) in comparison to H<sub>2</sub> [7,8]. As shown in Fig. 1a, theoretical  $\epsilon_{\text{RT,max}}$  for the H<sub>2</sub>-based energy storage decreases nearly linearly with increasing temperature, reaching 76% at 800 °C near the operating temperature of SOECs. In contrast, theoretical  $\epsilon_{\text{RT,max}}$  is nearly 100% and temperature-invariant for CH<sub>4</sub>-based energy storage. The wide, high-efficiency operating window also brings a measure of system robustness.

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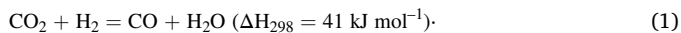
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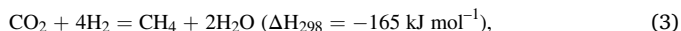
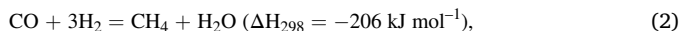
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Methane production through co-electrolysis of  $\text{CO}_2$  and  $\text{H}_2\text{O}$  has been demonstrated with oxygen-ion-conducting SOECs (O-SOECs) [9–13]. During operation,  $\text{CO}_2$  and  $\text{H}_2\text{O}$  feedstocks are co-fed to the fuel electrode, where water vapor is electrolyzed and oxygen ions are driven across the electrolyte. This leaves  $\text{H}_2$  at the fuel electrode to facilitate  $\text{CO}_2$  reduction through the reverse water-gas shift reaction:



Methane is formed through  $\text{CO}/\text{CO}_2$  hydrogenation:

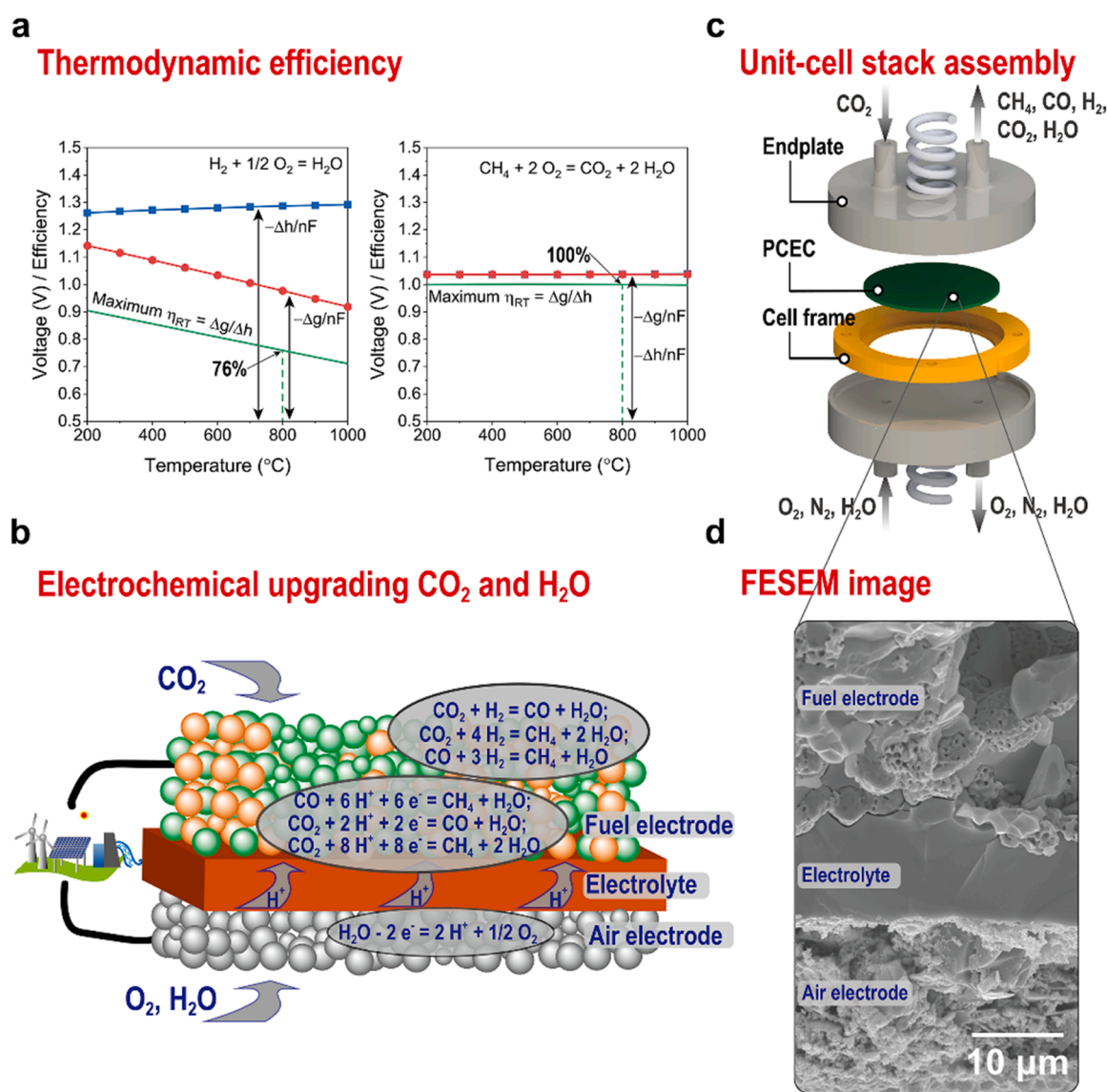


with Rx. (3) known as the Sabatier reaction. The integration of exothermic methane production with the endothermic water electrolysis in a single device enables in-situ heat balancing to boost conversion efficiency. Further, the integration simplifies the system, facilitates

thermal management, and improves reliability.

Li, et al. [9] attempted to achieve direct  $\text{CH}_4$  production utilizing an O-SOEC with conventional nickel/yttria-stabilized zirconia (Ni-YSZ) fuel electrode material set. With a feedstock of  $50 \text{ mL min}^{-1} \text{ H}_2\text{O}$  and  $25 \text{ mL min}^{-1} \text{ CO}_2$  to fuel electrode, a  $\text{CH}_4$ -yield ratio of 0.03% (where the  $\text{CH}_4$  yield ratio is defined as the  $\text{CH}_4$  production /  $\text{CO}_2$  feed) was obtained under a total electrolysis current of  $-0.608 \text{ A}$  at  $550^\circ\text{C}$ . Luo, et al. [11] reported a similar  $\text{CH}_4$ -yield ratio of 0.03% by using a tubular O-SOEC with the same Ni-YSZ fuel electrode at  $600^\circ\text{C}$  under a total electrolysis current of  $-0.97 \text{ A}$ . Featuring a novel  $\text{Fe-La}_{0.2}\text{Sr}_{0.8}\text{TiO}_{3-\delta}$  fuel electrode, Xie, et al. [10] demonstrated an improved  $\text{CH}_4$ -yield ratio of 0.2% at  $650^\circ\text{C}$  with a feedstock of  $2 \text{ mL min}^{-1} \text{ H}_2\text{O}$  and  $1 \text{ mL min}^{-1} \text{ CO}_2$ .

These low  $\text{CH}_4$ -yield ratios are rooted in the typical high ( $700\text{--}900^\circ\text{C}$ ) operating temperatures of O-SOECs, at which the exothermic methanation reactions are not thermodynamically favored (Fig. S1a). Additionally, co-feeding  $\text{H}_2\text{O}$  with  $\text{CO}_2$  drives the methanation reaction towards the reactants through Le Chatelier's principle



**Fig. 1.** Illustration of PCEC unit-cell stack for  $\text{CO}_2$ - $\text{H}_2\text{O}$  co-conversion. a, Comparison of theoretical thermodynamic cell-level maximum round-trip efficiency,  $\varepsilon_{RT,max}$ , between  $\text{H}_2$ -based (left) and  $\text{CH}_4$ -based (right) energy storage options. b, Schematic illustration of the possible electrochemical and chemical reactions in the PCEC during upgrading  $\text{CO}_2$ - $\text{H}_2\text{O}$  into  $\text{CH}_4$ .  $\text{CO}_2$  reduction may occur via the direct electrochemical pathway where adsorbed protons and electrons reduce  $\text{CO}_2$  directly, or by the indirect pathway where  $\text{CO}_2$  is thermochemically hydrogenated by electrochemically generated  $\text{H}_2$ . c, Unit-cell stack assembly of the PCEC. The cell is bonded into a ceramic frame and then sandwiched by two ferritic-steel endplates. (See Fig. 2 for detailed pictures of the assembly) d, Cross-sectional morphology of a freshly prepared and reduced PCEC.

(Fig. S1c). To facilitate CH<sub>4</sub> production, supplying additional H<sub>2</sub> has proved to be effective [9,11], in agreement with thermodynamic predictions (Fig. S1b). For instance, Luo, et al. [11] improved the CH<sub>4</sub>-yield ratio from 0.03% to 12% by switching the reactant to a 1:1:1 gas mixture of CO<sub>2</sub>-H<sub>2</sub>O-H<sub>2</sub> while keeping other operating conditions the same. Further, Chen, et al. [12] and Lei, et al. [13] exploited the tubular cell architecture to create two distinct high- and low-temperature zones along the O-SOEC, with the former for high-efficiency H<sub>2</sub>O electrolysis, and the latter for high-selectivity methanation. Combined with supplemental H<sub>2</sub>, these innovations increased the CH<sub>4</sub>-yield ratio to 41% [12] and 84% [13], respectively. However, the split-zone approach engenders thermo-mechanical stresses to the delicate electro-ceramic cell components and brings reliability concerns. Co-feeding extra H<sub>2</sub>, the amount of which cannot be fulfilled by recycling exhaust H<sub>2</sub> [11–13], also requires substantial cost and energy input for hydrogen production.

Alternatively, PCECs recently emerge as a novel type of SOEC for the production of chemical fuels [14–19]. During water electrolysis, Duan, et al. [14], Choi, et al. [15] and Vollestad, et al. [16] have separately attested hundreds to thousands of hours of stable operation with high Faradaic efficiencies, with scale-up to multi-cell stacks now underway [17]. PCECs also offer several important benefits integral to achieving high-efficiency direct CH<sub>4</sub> production. The small size of the protonic charge carrier enables higher ionic conductivity at lower operating temperatures (400–600 °C) in comparison to O-SOECs. These lower temperatures present a better balance between the thermodynamics of the Sabatier chemistry favored at a lower temperature, and the chemical kinetics of CO<sub>2</sub> conversion and CH<sub>4</sub> formation favored at a higher temperature [20]. Most significantly, protonic ceramics provide physical separation of the CO<sub>2</sub> and H<sub>2</sub>O feedstocks at opposing electrodes, as illustrated in Fig. 1b. Higher CO<sub>2</sub> conversions have been demonstrated in sorption-enhanced [21] and membrane reactors [22], where timely removal/adsorption of product water shifts the thermodynamic equilibrium of methanation reaction towards the products. The physical separation in the PCEC presents a similar condition; protons formed through H<sub>2</sub>O electrolysis at the air electrode serve as the H<sub>2</sub> source for CO<sub>2</sub> hydrogenation at the fuel electrode, eliminating the existence of reactant H<sub>2</sub>O in the fuel electrode and facilitating higher CH<sub>4</sub> production.

Researchers are now harnessing protonic ceramics to directly produce CH<sub>4</sub> from CO<sub>2</sub> and H<sub>2</sub>O, with significant improvement on the CH<sub>4</sub>-yield ratio observed. Xie, et al. [23] employed a protonic ceramic membrane as a hydrogen pump to produce CH<sub>4</sub> from CO<sub>2</sub> and H<sub>2</sub> with a yield ratio of 1.4% using a fuel electrode composed of Fe and BaCe<sub>0.5</sub>Zr<sub>0.3</sub>Y<sub>0.16</sub>Ni<sub>0.04</sub>O<sub>3-δ</sub> at 614 °C. Li, et al. [24] also demonstrated a thermo-electrochemical production of C<sub>1</sub> species with a similar approach from H<sub>2</sub> and CO<sub>2</sub>. By tuning of the Ir–O hybridization of the Ir–ceria-based catalysts, a CH<sub>4</sub> selectivity of nearly 100% was achieved. Duan, et al. [14] demonstrated the direct synthesis of CH<sub>4</sub> from only CO<sub>2</sub> and H<sub>2</sub>O reactants using a small (0.5 cm<sup>2</sup> active area) PCEC “button cell”, reaching a CH<sub>4</sub>-yield ratio of 7.5% at 500 °C with a total electrolysis current of –1.625 A. While this is much higher than the previous 0.2% CH<sub>4</sub>-yield ratio obtained from O-SOECs, still higher yield ratios are needed for commercialization.

The fundamental electrochemical and thermochemical processes underway during protonic-ceramic electrolysis of CO<sub>2</sub> are also unclear. As shown in Fig. 1b, the potential role of direct electrochemical CO<sub>2</sub> reduction from protons, and its interplay with thermochemical CO<sub>2</sub> reduction from electrochemically produced H<sub>2</sub>, warrant further study. The difference is important as the electric field could possibly promote the reaction, which is referred to as the electrochemical promotion of the catalysis (EPOC) effect [25–28]. Jiménez, et al. [26] found that with Ni or Ru catalyst supported on YSZ, the CH<sub>4</sub> formation rate during CO<sub>2</sub> hydrogenation increased from  $0.90 \times 10^{-8}$  mol s<sup>–1</sup> to  $0.99 \times 10^{-8}$  mol s<sup>–1</sup> upon the application of a negative potential of –1.30 V. Recently, Zagoriais, et al. [27] and Kalaitzidou, et al. [28] have also observed promotion of the CO<sub>2</sub> hydrogenation process with Ru

nanoparticles supported on proton-conducting yttria-doped barium zirconate electrolyte under bias. However, to the authors’ knowledge, direct comparison between electrochemical and thermochemical CO<sub>2</sub> upgrade in PCECs during CO<sub>2</sub>-H<sub>2</sub>O co-conversion has not been quantitatively investigated.

Motivated by the potential operational benefits brought by protonic ceramics, here we successfully fabricated and assembled larger-area (5 cm<sup>2</sup> active area) PCEC unit-cell stacks (Fig. 1c). Comparing with previous lab-scale button cell study, the unit-cell stack as developed, composed of one cell and two endplates, enables more-efficient utilization of reactants and the evaluation of the performance from a broader system-level viewpoint in practical application. The fuel-cell-mode operation with humidified CH<sub>4</sub> was first conducted to attest the scalability of the protonic ceramic cells. Afterwards, direct CH<sub>4</sub> production through electrochemical CO<sub>2</sub>-H<sub>2</sub>O co-conversion was characterized and promising CH<sub>4</sub>-yield ratios were demonstrated. Additionally, a direct comparison was made between the electrochemical conversion of CO<sub>2</sub>-H<sub>2</sub>O and the thermochemical conversion of CO<sub>2</sub>-H<sub>2</sub> using the same cell. We find that in some conditions, this electrochemical CH<sub>4</sub>-synthesis route gives a higher CH<sub>4</sub> yield than the thermochemical route, demonstrating a possible EPOC effect. Lastly, techno-economic analyses were conducted to identify system-scale operating conditions that present an encouraging leveled cost of fuel production.

## 2. Experimental

### 2.1. Fabrication of 5-cm<sup>2</sup> PCECs

The fuel electrode-supported planar PCECs were prepared following the solid-state reactive sintering (SSRS) process [29]. Detailed powder synthesis and cell fabrication processes can be found in the [Supplemental Material](#) and the cross-sectional image of a freshly prepared and reduced cell is shown in Fig. 1d. The morphology was examined using Field Emission Scanning Electron Microscopy (FESEM, JSM-7000F). The cell has a BaCe<sub>0.4</sub>Zr<sub>0.4</sub>Y<sub>0.1</sub>Yb<sub>0.1</sub>O<sub>3-δ</sub> (BCZYYb) electrolyte with a thickness of ~8 μm. The ~0.8-mm-thick fuel electrode is composed of 60 wt % NiO and 40 wt% BCZYYb together with an additional 20 wt% starch. The air electrode is made of 80 wt% BaCo<sub>0.4</sub>Fe<sub>0.4</sub>Zr<sub>0.1</sub>Y<sub>0.1</sub>O<sub>3-δ</sub> (BCFZY) and 20 wt% BCZYYb.

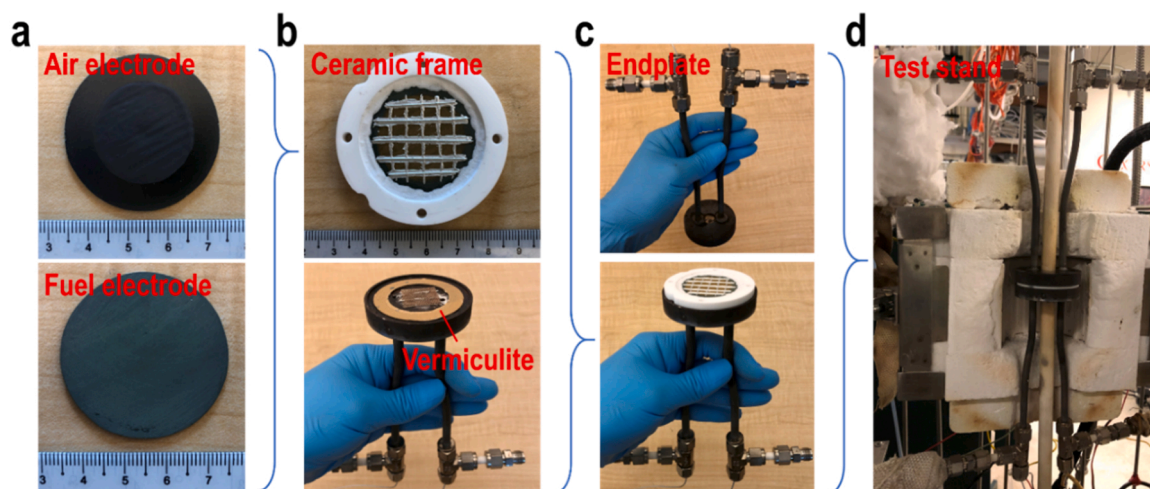
### 2.2. Assembly of a 5-cm<sup>2</sup> PCEC unit-cell stack

The schematic diagram of the unit-cell stack is presented in Fig. 1c and the photographs can be found in Fig. 2. The PCEC cell (Fig. 2a) was bonded into a ceramic frame with fuel electrode facing up and sealed using a ceramic bond paste (Ceramabond 552, Aremco Products Inc.) as shown in Fig. 2b top. The frame was then sandwiched by two identical metal endplates (Fig. 2c), made of Crofer 22 APU, and the sealing was achieved by vermiculite under compression force applied from a hydraulic jack (not shown) installed on the test stand (Fig. 2d). The gap between the fuel electrode and the endplate is ~2.5 mm in the assembly. The current collection was achieved via silver wires and paste (Fig. 2b top). The dimensions of the ceramic frame and the endplate are shown in Fig. S2.

### 2.3. Performance characterization of PCEC unit-cell stack

Electrochemical characterization was performed on a custom electrochemical test stand. The desired gas composition was achieved by regulating the supply of gases from individual gas cylinders using mass flow controllers. A heated bubbler was used to provide the desired steam flow by adjusting the set temperature. Current–voltage (*I*–*V*) and electrochemical impedance spectroscopy (EIS) measurements were performed using a Gamry Interface 5000E. EIS was carried out over a frequency ranging from 100 kHz to 0.1 Hz with an AC amplitude of 10 mV under open-circuit voltage (OCV).





**Fig. 2.** Pictures of the PCEC unit-cell stack and the test station. a, the as-prepared PCEC cell; b and c, the unit-cell stack assembly; d, the test stand (The compression force was applied from a hydraulic jack (not shown) on the cell with the two alumina tubes).

In direct-CH<sub>4</sub>-fueled fuel-cell mode, the *I*–*V* curves were obtained across a range of temperatures with a mixture of 20 sccm CH<sub>4</sub>, 80 sccm N<sub>2</sub> and 67 sccm steam as the fuel to produce a steam-to-carbon ratio = 3.35. The high N<sub>2</sub> flow rate is to reduce the steam concentration and minimize fluctuations in steam flow rate, but likely to decrease the cell performance due to the fuel-dilution effect.

During electrochemical CO<sub>2</sub>–H<sub>2</sub>O co-conversion, the air electrode was supplied with 200 sccm synthetic air with 20% steam and the fuel electrode was fed with 48 sccm N<sub>2</sub> and 2 sccm CO<sub>2</sub>. The 4% CO<sub>2</sub> concentration in the stream was selected to be consistent with previous tests on button cells [14]. This feedstock results in H<sub>2</sub>-to-CO<sub>2</sub> ratios ranging from ~ 2:1 to ~ 6:1 in the fuel electrode stream depending on the electrolysis current density. Current densities varied from – 0.4 A cm<sup>–2</sup> to – 1 A cm<sup>–2</sup> with an interval of 0.1 A cm<sup>–2</sup> were applied and kept for 1 h for each step. The fuel-electrode exhaust gas flow was monitored by a gas flow calibrator (Defender 530 +, Mesa Labs) and a gas chromatograph (3000 Micro GC, Agilent). Any steam content in the exhaust gas flow was removed by calcium sulfate desiccant (W. A. Hammond DRIERITE Co. LTD) prior to the gas flow calibration and gas chromatography. Calculation of Faradaic efficiencies (FEs) is detailed in the [Supplemental Material](#). The Faradaic efficiency towards all the products is denoted as FE<sub>total</sub>, while FE<sub>CH<sub>4</sub></sub> indicates the Faradaic efficiency towards CH<sub>4</sub> production.

The thermochemical CH<sub>4</sub> synthesis was conducted by co-feeding H<sub>2</sub> and CO<sub>2</sub> into the fuel-electrode chamber while keeping all other operating parameters the same as those used in electrochemical CH<sub>4</sub> synthesis. The flow rate of CO<sub>2</sub> and N<sub>2</sub> were kept at 2 sccm and 48 sccm, respectively, while the flow rate of H<sub>2</sub> was varied.

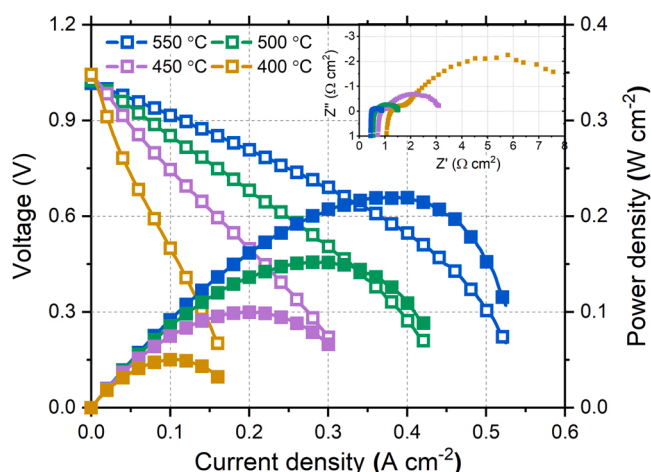
#### 2.4. Techno-economic analysis

To identify system-scale operating conditions, the economic performance of the electrochemical CO<sub>2</sub> upgrade with PCECs was evaluated using a levelized cost of fuel production (LCOFP) analysis. Detailed analysis procedures can be found in the [Supplemental Material](#).

### 3. Results and discussion

#### 3.1. PCEC performance in fuel-cell mode with humidified CH<sub>4</sub>

The electrochemical performance of the unit-cell stack was first characterized in fuel-cell mode with humidified CH<sub>4</sub> fuel (Fig. 3). The open-circuit voltage (OCV) ranged from 1.02 (at 550 °C) to 1.04 V (at 400 °C), indicating good quality of the large-area cell and adequate gas

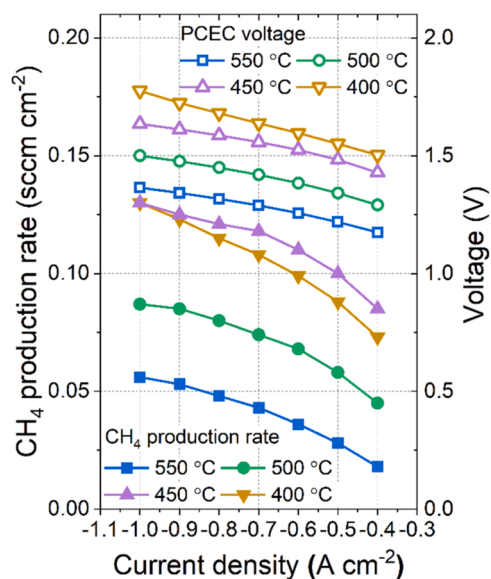


**Fig. 3.** Current-voltage and current-power characterization of the protonic ceramic unit-cell stack in CH<sub>4</sub>-fueled fuel-cell mode with electrochemical impedance spectra shown in the inset.

sealing within the stack package. Peak power densities were demonstrated to be 0.22, 0.15, 0.10 and 0.05 W cm<sup>–2</sup> at 550, 500, 450 and 400 °C, respectively. Higher performance can be obtained through a higher-temperature operation that facilitates CH<sub>4</sub> reforming and electrochemical reaction [30]. Also, as mentioned earlier, the high N<sub>2</sub> concentration used here likely decreased the performance due to fuel-dilution effects. The area-specific ohmic and polarization resistances at 550 °C reached 0.48 and 0.41 Ω cm<sup>2</sup>. These results are comparable with previously demonstrated state-of-the-art performances for small protonic ceramic button cells (0.5 cm<sup>2</sup> active area) [29,31], though the electrochemically active area has been increased in this case by an order of magnitude. The larger cell size and packaging within a conventional SOFC-stack material set demonstrates an encouraging measure of device scalability.

#### 3.2. PCEC performance in electrolysis mode for CO<sub>2</sub>–H<sub>2</sub>O co-conversion

Electrochemical polarization curves and methane-production rates during CO<sub>2</sub>–H<sub>2</sub>O co-conversion are shown in Fig. 4. In order to achieve an electrolysis current density of –1 A cm<sup>–2</sup>, the applied voltages were 1.36, 1.50, 1.63 and 1.78 V at 550, 500, 450 and 400 °C, respectively. Peak CH<sub>4</sub> production was achieved at 450 °C under –1 A cm<sup>–2</sup> with a



**Fig. 4.** Characteristic current–voltage curves and methane production rates for CO<sub>2</sub>–H<sub>2</sub>O co-conversion with the PCEC unit-cell stack as a function of current density at different temperatures. The flow rate of CO<sub>2</sub> to the fuel-electrode chamber is 2 sccm, carried by 48 sccm N<sub>2</sub>.

production rate of 0.13 sccm cm<sup>−2</sup>. The non-linear increase of the methane production rate with increasing current density is in agreement with the thermodynamic predictions for the CH<sub>4</sub> yield with an increasing H<sub>2</sub>-to-CO<sub>2</sub> ratio (Fig. S1b). Methane production is also highly temperature sensitive; CH<sub>4</sub> production at 400–450 °C is more than double the value found at 550 °C. Further, the CH<sub>4</sub> formation is steady between 400 and 450 °C, presenting a wide and robust operating window. This result experimentally validates the potential application of protonic-ceramic electrolyzers for CO<sub>2</sub> upgrading into practical fuels.

### 3.3. Performance metrics of in-situ CH<sub>4</sub> production during CO<sub>2</sub>–H<sub>2</sub>O co-conversion

Fig. 5 shows the evolution of the cell voltage and product-gas composition with a stepwise increase in the applied current density during CO<sub>2</sub>–H<sub>2</sub>O co-conversion. Both the cell voltage and the CH<sub>4</sub>, CO and H<sub>2</sub> product yields are reasonably stable at each current density. The ratios of carbon out to carbon in were calculated and the result is shown in Fig. S3. Mostly, the ratio falls in the range of 92–95%, which could be due to the fact that other possible products such as C<sub>2+</sub> fuels were not monitored. Further, no morphological damages can be observed from post-mortem analysis of PCEC electron micrographs (Fig. S4). These observations establish at least the short-term stability of the cell.

The CH<sub>4</sub> production performance metrics, CO<sub>2</sub> conversion, CH<sub>4</sub> selectivity and CH<sub>4</sub>-yield ratio, are shown in Fig. 6. Because of the exothermic nature of CO<sub>2</sub> methanation, lower operating temperatures are thermodynamically favorable but kinetically unfavorable. As shown in Fig. 6a and b, this interplay leads to increasing CO<sub>2</sub> conversion but decreasing CH<sub>4</sub> selectivity with increasing temperature. Reflecting the tradeoff between these two factors, the CH<sub>4</sub>-yield ratio increased first from 14.5% to 34.6% when the operating temperature decreased from 550 °C to 450 °C, and then decreased to 32.7% at 400 °C (Fig. 6c), with an electrolysis current of −1 A cm<sup>−2</sup>. Overall, the CH<sub>4</sub>-yield ratio is maximized to 34.6% at an intermediate temperature of 450 °C; to the authors' knowledge, this is the highest electrochemical CH<sub>4</sub>-yield ratio from CO<sub>2</sub>–H<sub>2</sub>O co-conversion reported to date. Apart from the lower operating temperature of the PCEC, the high CH<sub>4</sub>-yield ratios are also a result of the physical separation of the CO<sub>2</sub> and H<sub>2</sub>O feedstocks at opposing electrodes. Fig. S5 compares the performance metrics of

CO<sub>2</sub>–H<sub>2</sub>O co-conversion with and without additional 5% steam in the gas stream supplied to the fuel electrode. The CH<sub>4</sub>-yield ratio decreases from 16.8% to 6.6% under a current density of −1 A cm<sup>−2</sup> at 550 °C upon the addition of steam. This result agrees with our thermodynamic predictions (Fig. S1c) and previous reports [21,22]. Lastly, the CH<sub>4</sub>-yield ratio increased with the increasing current density, as a result of increasing H<sub>2</sub>-to-CO<sub>2</sub> ratio. We hypothesize that higher current densities could further facilitate CH<sub>4</sub> production.

Fig. 6d demonstrates the change of CO<sub>2</sub> conversion, CH<sub>4</sub> selectivity and CH<sub>4</sub>-yield ratio with the increase of the equivalent H<sub>2</sub>-to-CO<sub>2</sub> ratio at 450 °C. Equivalent H<sub>2</sub>-to-CO<sub>2</sub> ratios were calculated by dividing the equivalent H<sub>2</sub> flow rate in the exhaust stream,  $\dot{V}_{H_2,eq}$  (Eq. S2), by the flow rate of input CO<sub>2</sub>. When the equivalent H<sub>2</sub>-to-CO<sub>2</sub> ratio increased from 3.14 to 5.35 by imposing higher current densities, the CH<sub>4</sub>-yield ratio increased from 23.2% to 34.6%. Previously, the CH<sub>4</sub>-yield ratio during CO<sub>2</sub>–H<sub>2</sub>O co-conversion was reported to be 7.5% with an equivalent H<sub>2</sub>-to-CO<sub>2</sub> ratio of around 5: 1 [14]. In comparison, the CH<sub>4</sub>-yield ratio reached 33.7% in this work under the same H<sub>2</sub>-to-CO<sub>2</sub> ratio, demonstrating the promising applicability of PCEC in electrochemical CO<sub>2</sub> upgrade with a practical stack assembly.

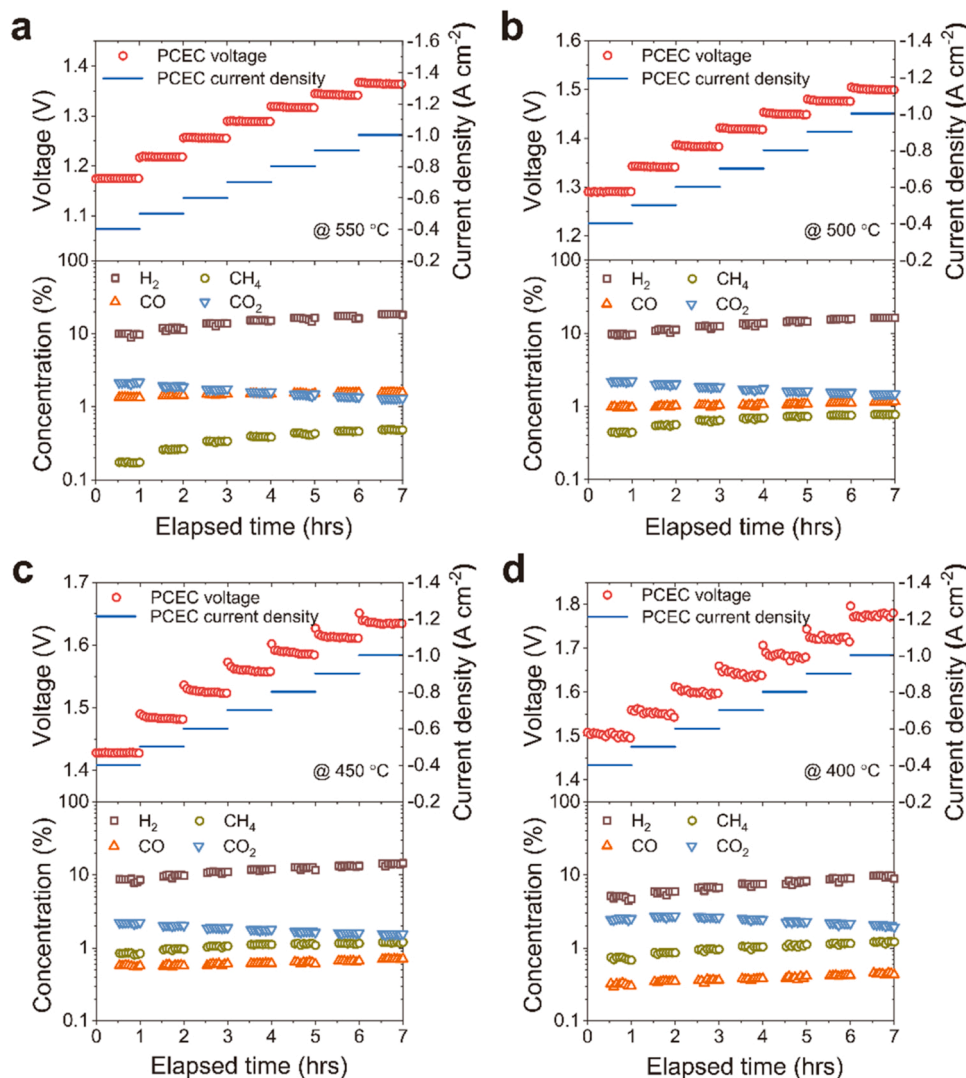
The FEs of the cell during CO<sub>2</sub>–H<sub>2</sub>O co-conversion are shown in Fig. S6. Protonic-ceramic materials are mixed ionic-electronic conductors. The incorporation of molecular oxygen into oxygen vacancies in oxidizing environments will result in the formation of electron holes [32–34]. This leads to non-negligible electronic conductivity and reduces the transference number of protons. Therefore, the actual protonic current density in the electrolyte is smaller than the applied external electrolysis current density. In short, the formation of electron holes negatively affects the FE. In proton-conducting BCZYB ceramic, the concentration of electron holes can be related to operating conditions, such as oxygen and steam partial pressures, through the water oxidation equilibrium via [16,35]:

$$[h^{\cdot}] = K [OH^{\cdot}] [O^{\cdot}] p_{H_2O}^{-\frac{1}{2}} p_{O_2}^{\frac{1}{4}}, \quad (4)$$

assuming full hydration. The FE<sub>total</sub> was calculated to be in the range of 20%–50% (Fig. S6a); this is reflective of the relatively oxidizing nature of these two feedstocks. During water electrolysis, researchers have demonstrated high FEs [14–16]. Further efforts will be spent on improving the FEs during CO<sub>2</sub>–H<sub>2</sub>O co-conversion. The FE<sub>CH<sub>4</sub></sub> is shown in Fig. S6b. A maximum value of 12% was obtained at 450 °C, consistent with the optimal operating temperature for CH<sub>4</sub> synthesis.

Generally, the FE<sub>total</sub> decreased with increasing electrolysis current density. This could be a result of the consumption of steam and the production of oxygen at higher electrolysis current densities, which lowers the local  $p_{H_2O}$  and increases the local  $p_{O_2}$  at the air electrode. Further, the FE<sub>total</sub> decreased with decreasing operating temperature. This is in contrast to the theoretical predictions based on the transportation properties of protonic ceramics [33,34,36]. However, the polarizations occurring at the electrodes and the choices of both the electrolyte and electrodes may greatly affect the cell performance [14–16]. More efforts are needed to comprehensively investigate the transportation behaviors of protonic ceramics during CO<sub>2</sub>–H<sub>2</sub>O co-conversion.

Further attempts were made to boost the CH<sub>4</sub>-yield ratio. As a first step in this direction, we achieved an improvement on CH<sub>4</sub>-yield ratio from 34.6% to 40.9% at 450 °C (Fig. S7a) by increasing the electrolyte thickness from ~8 to ~12 μm. Through modeling approach, Nakamura, et al. [37] and Qiu, et al. [38] have demonstrated that increasing the electrolyte thickness can suppress leakage current and is beneficial for the improvement of the FE in the fuel-cell mode. Zhang, et al. [39] also experimentally demonstrated that a thicker electrolyte would lead to a higher open-circuit voltage. Considering the strong oxidizing environment at the air electrode, the electron-hole formation mainly takes place at the air electrode side of the electrolyte. Due to the thin electrolyte used, it is speculated that the hole formation dominates the



**Fig. 5.** Evolution of voltage and exhaust gas concentrations with elapsed time of the PCEC during  $\text{CO}_2\text{--H}_2\text{O}$  co-conversion at different temperatures. Stable voltage and exit gas composition at each current density demonstrate the short-term stability of the cell. Electron micrographs of the cell before and after the operation shown in Fig. S4 reveal no morphological damage.

electron-hole concentration profile. Thus, increasing the thickness of the electrolyte while keeping other parameters the same will lower the electron-hole concentration at the fuel-electrode side and improve the FE. Fig. S8 shows the EIS of the two cells tested at 550 °C. The ohmic resistance increased from  $0.31 \Omega \text{ cm}^2$  to  $0.44 \Omega \text{ cm}^2$  as a result of the thicker electrolyte. Consequently, Fig. S7c shows that the  $\text{FE}_{\text{total}}$  was improved from 30.6% to 39.8% at a current density of  $-1 \text{ A cm}^{-2}$ , leading to higher  $\text{H}_2$  production rates and thus a higher  $\text{CH}_4$ -yield ratio. Meanwhile, the cell voltage was increased by  $\sim 4\%$ , from 1.63 V to 1.72 V (Fig. S7b), indicative of the increased DC resistance brought with the thicker electrolyte. Nonetheless, the effect of electrolyte thickness on the FE warrants further detailed investigation.

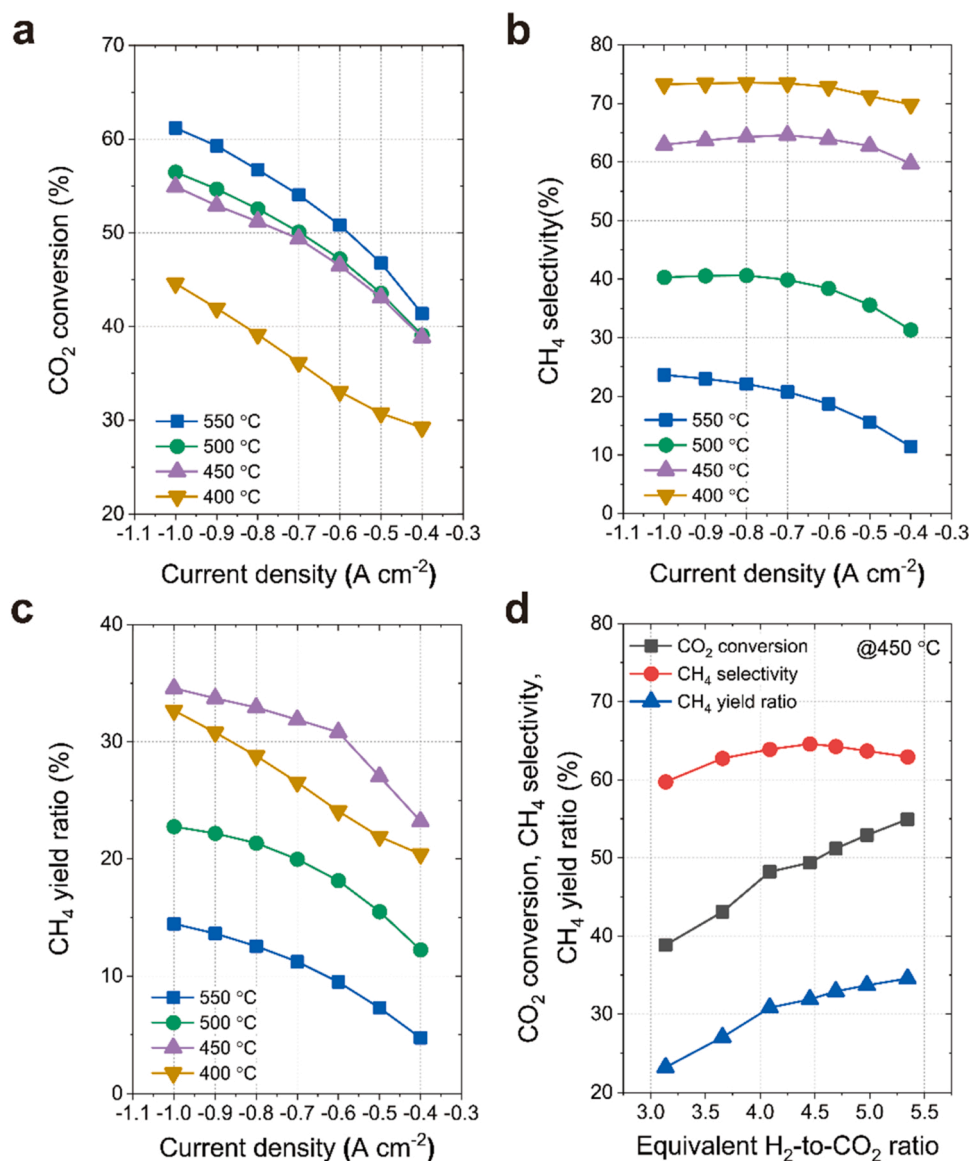
Alternatively, increasing the  $\text{H}_2$ -to- $\text{CO}_2$  ratios by recycling exhaust  $\text{H}_2$  to the fuel electrode together with the  $\text{CO}_2$  feed enables a significant increase in the  $\text{CH}_4$ -yield ratio, from 34.6% to 71.2% at 450 °C with an electrolysis current of  $-1 \text{ A cm}^{-2}$  (Fig. 7a), marking a 106% improvement. In the experiment, supplemental  $\text{H}_2$  with a flow rate lower than the flow rate of  $\text{H}_2$  in the exhaust gas stream (Fig. 7c) was provided. Thus, assuming that efficient downstream separation of  $\text{CH}_4$  and  $\text{H}_2$  can be executed, the supplemental  $\text{H}_2$  demand can be met by recycling electrochemically produced  $\text{H}_2$ , without the need for external  $\text{H}_2$  supply. The supply of recycled  $\text{H}_2$  also suppresses hole formation [33], and

therefore improves the  $\text{FE}_{\text{total}}$  of the stack under the same current density (Fig. 7b). A comparison between the performance with and without recycling exhaust  $\text{H}_2$  against equivalent  $\text{H}_2$ -to- $\text{CO}_2$  ratio is shown in Fig. 7d. It can be seen that the  $\text{CH}_4$ -yield ratios show higher values with  $\text{H}_2$  recycle, benefitting from the parallel routes of thermochemical and electrochemical  $\text{CO}_2$  methanation. These effects synergistically improve the stack energy density and energy efficiency. As will be shown, the performance improvements brought by  $\text{H}_2$  recycle also substantially decrease the LCOFP.

Overall, in a practical stack configuration, the PCEC enables a  $\text{CH}_4$ -yield ratio of 34.6% without  $\text{H}_2$  recycle and 71.2% with  $\text{H}_2$  recycle. As found in Table S1–S2, these values exceed the previously reported values of 0.3–7.5% from direct  $\text{CO}_2$  and  $\text{H}_2\text{O}$  co-conversion using button cells [9,10,14,23], and are comparable to previous values ranging from 12% to 84% with external  $\text{H}_2$  supply using button cells or with two distinct temperature zones using tubular cells [3,9,11–13]. The much-improved  $\text{CH}_4$ -yields presented here compared with the previous lab-scale button-cell tests demonstrate a promising result for future upscale and practical application of PCECs.

Compared with  $\text{CO}_2$  reduction and  $\text{CH}_4$  production in the aqueous system, the work here shows a higher areal  $\text{CH}_4$  production rate, while maintaining a similar  $\text{FE}_{\text{CH}_4}$  (Table S3). The highest  $\text{CH}_4$  production rate





**Fig. 6.** CO<sub>2</sub> conversion metrics of the PCEC during CO<sub>2</sub>-H<sub>2</sub>O co-conversion. a–c, Dependence of CO<sub>2</sub> conversion, CH<sub>4</sub> selectivity and CH<sub>4</sub> yield ratio on the current density at different temperatures. d, Dependence of CO<sub>2</sub> conversion, CH<sub>4</sub> selectivity and CH<sub>4</sub> yield on the equivalent H<sub>2</sub> ( $\dot{V}_{\text{H}_2, \text{eq}} = \dot{V}_{\text{H}_2} + \dot{V}_{\text{CO}} + 4 \times \dot{V}_{\text{CH}_4}$ ) to CO<sub>2</sub> ratio at 450 °C.

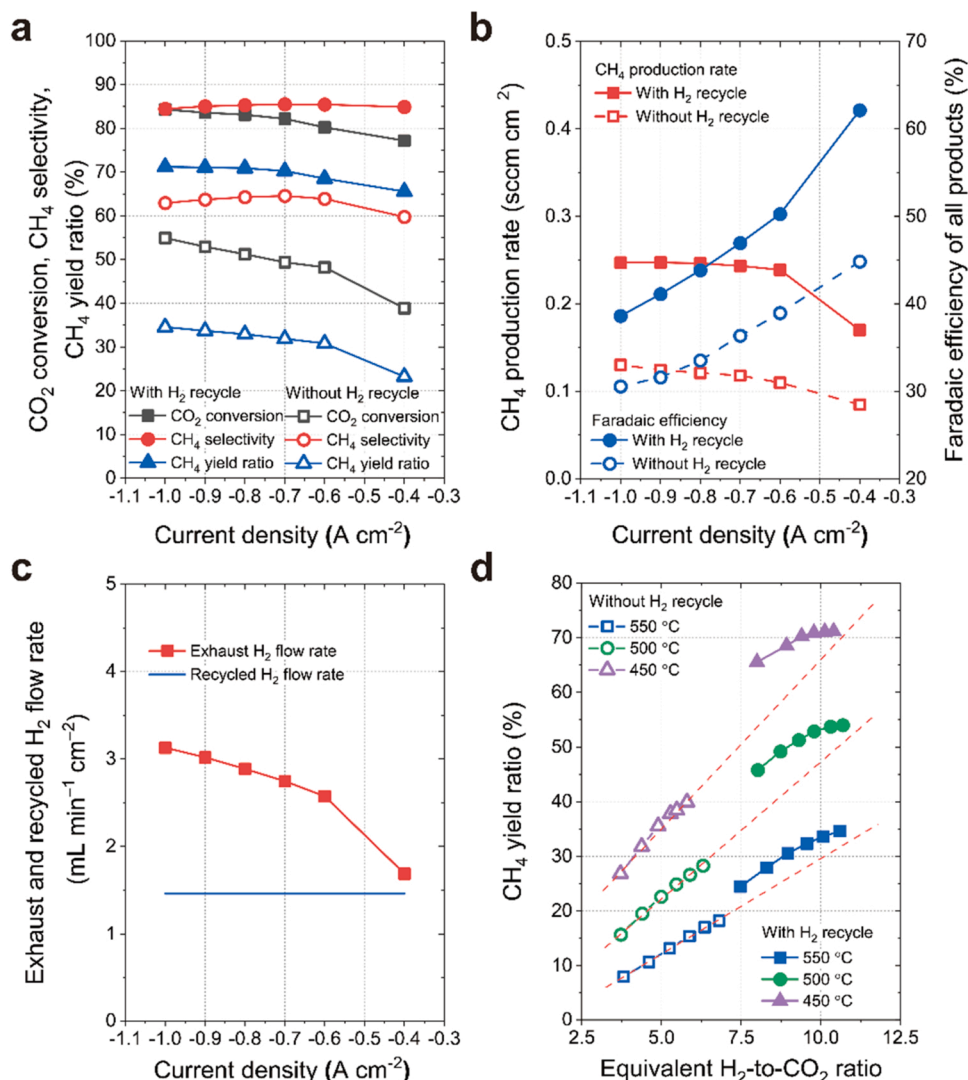
achieved here is 0.13 sccm cm<sup>-2</sup> without H<sub>2</sub> recycle and 0.25 sccm cm<sup>-2</sup> with H<sub>2</sub> recycle, which translates to CH<sub>4</sub> production rates of 97 nmol s<sup>-1</sup> cm<sup>-2</sup> and 180 nmol s<sup>-1</sup> cm<sup>-2</sup>, respectively. In comparison, CH<sub>4</sub> production rates were reported to be ~ 0.05 nmol s<sup>-1</sup> cm<sup>-2</sup> using metal-nitrogen-doped carbon catalysts [40], 57 nmol s<sup>-1</sup> cm<sup>-2</sup> using a cobalt phthalocyanine and zinc-nitrogen-carbon tandem catalyst [41], and 3.9 nmol s<sup>-1</sup> cm<sup>-2</sup> with a polycrystalline Cu catalyst [42]. The higher CH<sub>4</sub> production rates provided by the PCEC directly improve system energy density to reduce capital costs, as can be seen in the techno-economic analysis presented later.

### 3.4. Comparison between electrochemical and thermochemical CH<sub>4</sub> production

The emergence of protons from the electrolyte into the CO<sub>2</sub>-rich fuel electrode brings questions about the participation of activated H<sup>+</sup>, in parallel with H<sub>2</sub>, in the CO<sub>2</sub>-to-CH<sub>4</sub> reaction pathway. To explore this further, we compared the co-conversion performance under electrochemically produced H<sub>2</sub> with that where the equivalent H<sub>2</sub> is co-fed with

the CO<sub>2</sub> feedstock in the absence of an electrochemical driving current. The second approach treats the PCEC unit-cell stack as a fixed-bed reactor, where only thermochemically produced CH<sub>4</sub> is possible. The relative performances of the two CH<sub>4</sub>-synthesis paths are plotted against  $\dot{V}_{\text{H}_2, \text{eq}}$  and the results are shown in Fig. 8. It can be seen that the dependences of CO<sub>2</sub> conversion and CH<sub>4</sub> selectivity on the equivalent H<sub>2</sub> flow rate have altered. Perhaps most intriguingly, electrochemical synthesis results in higher overall CH<sub>4</sub> yields than chemical synthesis at operating temperatures  $\geq 450$  °C and low  $\dot{V}_{\text{H}_2, \text{eq}}$  (for example, < 9 sccm at 450 °C) (Fig. 8c). The comparison was performed again using the cell with thicker electrolyte at 550 °C and a similar improvement during electrochemical synthesis can be observed (Fig. S9). This means that with the same materials set and morphology, the electrochemical formation of hydrogen increases methane yield, marking the existence of a possible EPOC effect in electrochemical CO<sub>2</sub> upgrade using PCEC.

In gaseous catalytic reactions, the EPOC effect is universally observed in both single-chamber and two-chamber (electrochemical cell) configurations for a variety of reactions [43]. With an oxygen-ion conductor support, the origin of EPOC in the oxidation process has



**Fig. 7.** In-situ CH<sub>4</sub> production during CO<sub>2</sub>-H<sub>2</sub>O co-conversion using the PCEC unit-cell stack with H<sub>2</sub> recycle simulated by feeding additional H<sub>2</sub> to the fuel electrode together with CO<sub>2</sub> at 450 °C. a, Change of CO<sub>2</sub> conversion, CH<sub>4</sub> selectivity and CH<sub>4</sub>-yield ratio with current density; b, Change of CH<sub>4</sub> production rate and FE<sub>total</sub> with current density. The FE<sub>total</sub> is used instead of FE<sub>CH<sub>4</sub></sub>, as the formation of CH<sub>4</sub> contributed by the electrolyzed H<sub>2</sub> cannot be separated from the CH<sub>4</sub> production from recycled H<sub>2</sub>; c, Comparison of flow rates of exhaust H<sub>2</sub> and recycled H<sub>2</sub>, indicating that the additional H<sub>2</sub> feed can be fulfilled by exhaust H<sub>2</sub> recycle; d, Comparison of the CH<sub>4</sub>-yield ratio between CO<sub>2</sub> upgrade with and without H<sub>2</sub> recycle against equivalent H<sub>2</sub> (V<sub>H<sub>2</sub>,eq</sub> = V<sub>H<sub>2</sub></sub> + V<sub>CO</sub> + 4 × V<sub>CH<sub>4</sub></sub>) to CO<sub>2</sub> ratio (Red dash lines are to guide the eyes only).

been found to stem from the migration of anionic O<sup>δ-</sup> species from the support to the metal-gas interface, which alters the work function of the catalyst and serves as a sacrificial promoter to gas-phase O<sub>2</sub> reduction [44–47]. For the gaseous CO<sub>2</sub> hydrogenation process, catalysts supported with both oxygen-ion conducting electrolytes [26] and proton-conducting electrolytes [27,28] were demonstrated with electrochemically promoted catalytic performance. As in the thermochemical methanation of CO<sub>2</sub>/CO, the rate-limiting process is generally ascribed to either the dissociation of the C–O bond or the hydrogenation of the intermediates [48,49]. In O-SOECs, through surface chemistry characterizations via *operando* photoelectron spectroscopy [50,51], the CO<sub>2</sub> electrolysis was suggested to go through a CO<sub>3</sub><sup>2-</sup> intermediate. Upon anodic applied bias, the CO<sub>3</sub><sup>2-</sup> intermediate accumulates, suggesting the promotion of the pre-coordination of CO<sub>2</sub> to the catalyst surface [50]. In PCECs, Shi, et al. [52] also observed the formation of CO<sub>3</sub><sup>2-</sup> intermediates in the Ni-BZY fuel electrode, via in situ Raman and in situ diffuse reflectance infrared Fourier transform spectroscopy characterization. It was speculated that the continuously replenished protons play a critical role in the reduction of CO<sub>2</sub> in PCECs.

In this work, the CO<sub>2</sub> conversion was mainly promoted by the electrochemical route, while the CH<sub>4</sub> selectivity during electrochemical synthesis was equal to or lower than the thermochemical route, as revealed in Fig. 8a and b. Previously, Xie, et al. [23] also observed similar promotion on the CO<sub>2</sub> conversion with electrochemically

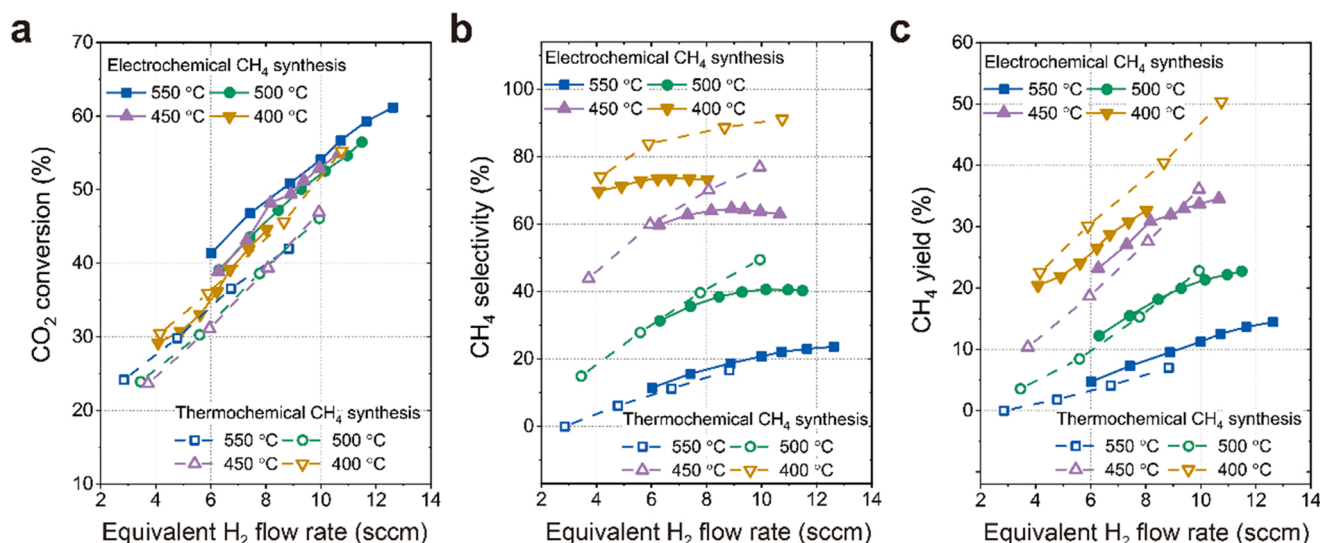
produced H<sub>2</sub> using the protonic ceramic membrane as a hydrogen pump. The enhanced CO<sub>2</sub> conversion indicates a promoted formation of CO compared to conventional thermochemical processes involving molecular H<sub>2</sub>. On the grounds of the above literature, two possible explanations can be raised. First, the applied bias and the protons may also facilitate the adsorption of CO<sub>2</sub> on the surface of Ni-BCZYb fuel electrode as proposed by Yu, et al. [50] and subsequently drive the conversion from CO<sub>3</sub><sup>2-</sup> to CO. From another perspective, the protons arriving at the fuel electrode might also work as a sacrificial promoter and directly participate in CO<sub>2</sub> reduction, schematically illustrated by the global hypothetical electrochemical CO<sub>2</sub> reduction (assuming a pure proton-conducting electrolyte):



Additionally, the electric field present during electrochemical CO<sub>2</sub> reduction may increase the electrochemical potential of the combined H<sup>+</sup> and e<sup>-</sup> and thus lower the transition state energy of the above reaction (Rx. 5). At lower temperatures and higher current densities, we hypothesize that H<sub>2</sub> evolution becomes increasingly favored and promotes the thermochemical route, while the EPOC effect brought by the applied bias and the protons is relatively less significant or even diminishing.

Admittedly, alternative possibilities such as contributions from partial oxygen-ion conductivity through the electrolyte or the different





**Fig. 8.** Comparison between electrochemical CH<sub>4</sub> synthesis and thermochemical CH<sub>4</sub> synthesis. Thermochemical CH<sub>4</sub> synthesis was examined by co-feeding CO<sub>2</sub> and H<sub>2</sub> into the fuel-electrode chamber of the PCEC unit-cell stack in the absence of electrochemically produced H<sub>2</sub>, using the stack as a fixed-bed reactor. All the panels are plotted as a function of the total equivalent H<sub>2</sub> flow rate ( $\dot{V}_{H_2,eq} = \dot{V}_{H_2} + \dot{V}_{CO} + 4 \times \dot{V}_{CH_4}$ ), in the exhaust gas. a–c, CO<sub>2</sub> conversion, CH<sub>4</sub> selectivity and CH<sub>4</sub> yield as a function of  $\dot{V}_{H_2,eq}$ . Generally, electrochemical CH<sub>4</sub> synthesis outperforms thermochemical CH<sub>4</sub> synthesis at high temperatures ( $\geq 450$  °C) and low  $\dot{V}_{H_2,eq}$  (for example,  $< 9$  sccm at 450 °C).

location of the H<sub>2</sub> source cannot be ruled out. Nonetheless, if an EPOC effect in PCECs can be validated, it would reinforce the potential benefits of integrating CH<sub>4</sub> synthesis and electrochemical CO<sub>2</sub>–H<sub>2</sub>O co-conversion in a single step. This would also suggest that similar proton-mediated electrochemical processes could potentially be harnessed to promote the activity of other hydrogenation reactions, such as the Haber–Bosch process for ammonia synthesis [53,54].

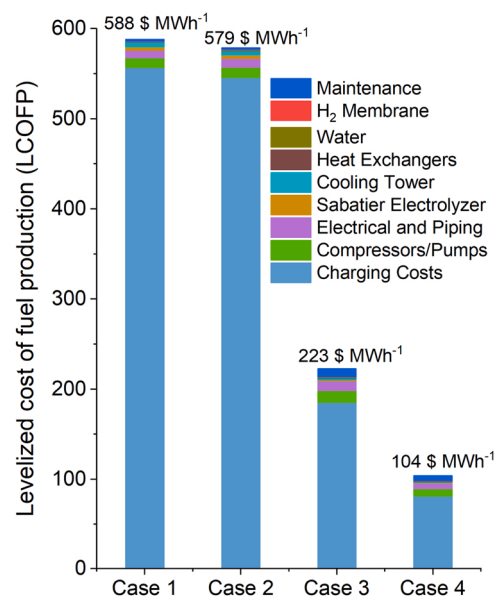
### 3.5. Techno-economic of CO<sub>2</sub> upgrading to CH<sub>4</sub>-based product gas with PCECs

We have pursued the techno-economic analysis of electrochemically upgrading CO<sub>2</sub> into CH<sub>4</sub>-based product gas to determine system-scale operating conditions and guide technological development. Levelized cost of fuel production (LCOFP) quantifies the required price-point of the product gas to enable “break-even” plant costs at the end of plant life (Eq. S4), presenting an insightful method for comparing the economics of various technologies. A PCEC-based plant model for CO<sub>2</sub> upgrade is shown in Fig. S10. LCOFP was compared across four operational state points detailed in Table S4. Detailed modeling process information can be found in the Supplemental Experimental Procedures. Economic assumptions can be found in Table S5.

LCOFPs for the four cases are compared in Fig. 9. Electricity cost is clearly the major contributor to total LCOFP, reflective of the energy required to co-convert the CO<sub>2</sub> and H<sub>2</sub>O into fuels. For the baseline case (Case 1, related to Fig. 4), all hydrogen is produced through H<sub>2</sub>O electrolysis; no H<sub>2</sub> recycle is used. The LCOFP is 588 \$<sub>2020</sub> MWh<sup>-1</sup>. The mildly oxidizing gas compositions bring low  $FE_{total}$  ( $\sim 30\%$ ), increasing both capital and operational costs.

Case 2 explores the impact of electrolyte thickness on LCOFP; the increased thickness boosts  $FE_{total}$  from 30% to 40%, but necessitates a higher driving voltage to achieve the  $-1$  A cm<sup>-2</sup> current density. Overall, increasing the electrolyte thickness only mildly reduces the LCOFP from 588 \$<sub>2020</sub> MWh<sup>-1</sup> to 579 \$<sub>2020</sub> MWh<sup>-1</sup>. The gains in the increasing CH<sub>4</sub>-yield ratio are partially offset by the higher power required.

The value proposition of H<sub>2</sub> recycle is captured in Case 3. LCOFP decreases substantially under H<sub>2</sub> recycle (Case 3), dropping to 223 \$<sub>2020</sub> MWh<sup>-1</sup>, less than half the baseline case. Recycling unconsumed H<sub>2</sub> could substantially improve the CH<sub>4</sub> production rate per unit area of the PCEC



**Fig. 9.** Levelized cost of fuel production (LCOFP) analysis of the proposed CO<sub>2</sub> upgrading plant based on PCEC technology. Detailed modeling process can be found in the Supplemental Experimental Procedures. Case 1 (Related to Fig. 4) and Case 2 (Related to Fig. S7, with thicker electrolyte) are without exhaust H<sub>2</sub> recycle, while Case 3 (Related to Fig. 7) is with exhaust H<sub>2</sub> recycle. The PCEC plant presents LCOFPs of 579 \$<sub>2020</sub> MWh<sup>-1</sup> without H<sub>2</sub> recycle (Case 2) and 223 \$<sub>2020</sub> MWh<sup>-1</sup> with H<sub>2</sub> recycle (Case 3). Further increasing the Faradaic efficiency to 90% and lowering the electricity cost to 0.02 \$ kWh<sup>-1</sup> (Case 4) lead to an LCOFP of 104 \$<sub>2020</sub> MWh<sup>-1</sup>.

stack (Fig. 7b). In addition, the more-reducing gas conditions brought by H<sub>2</sub> recycle greatly lower hole conduction, increasing the  $FE_{total}$ . The higher  $FE_{total}$  and increased areal CH<sub>4</sub> production rate enabled by H<sub>2</sub> recycle more than halves the electricity costs, and also reduces capital cost. Mass and energy balances for Case 3 can be found in Table S6. Cost contributions for Case 3 excluding charging cost are shown in Fig. S11.

The economics of fuel production rely heavily on economic factors,

such as electricity cost, and on technical indicators, such as the  $FE_{\text{total}}$ . Fig. S12a shows the cost breakdowns and the impact of  $FE_{\text{total}}$  on the LCOFP at a fixed electricity cost of 0.05 \$ kWh<sup>-1</sup>. PCEC performance with increasing  $FE_{\text{total}}$  was projected by holding the H<sub>2</sub>-to-CO<sub>2</sub> ratio, as well as other salient performance metrics – CO<sub>2</sub> conversion and CH<sub>4</sub> selectivity – constant. By increasing the  $FE_{\text{total}}$  from 62.1% to 100%, the LCOFP is reduced by 28% to 161 \$<sub>2020</sub> MWh<sup>-1</sup>. The impact of electricity cost on LCOFP is captured in Fig. S12b at a fixed  $FE_{\text{total}}$  of 62.1%. By decreasing the electricity cost from 0.05 \$ kWh<sup>-1</sup> to 0.01 \$ kWh<sup>-1</sup>, the LCOFP is further reduced by 58% to 93 \$<sub>2020</sub> MWh<sup>-1</sup>.

The synergistic effect of the electricity cost and the  $FE_{\text{total}}$  on LCOFP is shown in Fig. S13. When matching a low electricity cost of 0.02 \$ kWh<sup>-1</sup> with high  $FE_{\text{total}}$  of 90%, the LCOFP drops to 104 \$<sub>2020</sub> MWh<sup>-1</sup> (Fig. 9, Case 4). Such operating conditions are reasonable; in a previous report, Duan, et al. [14] demonstrated FE of over 90% during water electrolysis using a similar PCEC materials set. Further, the U.S. Department of Energy (DoE) has set a new goal of driving down the current cost of solar electricity to 0.02 \$ kWh<sup>-1</sup> by 2030 [55]. Thus, cost-effective CO<sub>2</sub> conversion to CH<sub>4</sub> is possible through a combination of advancements in renewables technologies and PCEC materials. Further materials advancements to reduce electronic conductivity in protonic ceramics and improve catalytic activity would clearly be of value.

#### 4. Conclusions

In this work, we have experimentally demonstrated a promising protonic-ceramic electrolysis cell for direct electrochemical co-conversion of CO<sub>2</sub>-H<sub>2</sub>O to methane. The important tradeoffs between operating conditions and performance metrics were explored from a broader viewpoint of system-level practicability using a PCEC unit-cell stack. Paths to significantly improve the CH<sub>4</sub>-yield ratio were demonstrated and the highest CH<sub>4</sub>-yield ratio of 34.6% by direct electrolytic co-conversion of CO<sub>2</sub>-H<sub>2</sub>O feedstocks was achieved at 450 °C. Co-feeding H<sub>2</sub> with CO<sub>2</sub> boosts the CH<sub>4</sub>-yield ratio to 71.2%. This supplemental hydrogen can be recycled from the reactor exhaust stream, where unused electrochemically produced H<sub>2</sub> can be harvested, eliminating a need for any external H<sub>2</sub> supply. In addition, we find that electrochemical CH<sub>4</sub> synthesis presents a possible EPOC phenomenon that results in a higher CH<sub>4</sub>-yield ratio than thermochemical CH<sub>4</sub> synthesis at high temperatures and low equivalent hydrogen flow rates. This could be ascribed to either faster CO<sub>2</sub> adsorption or work function alteration of the electrode by the protons acting as a sacrificial promoter under applied bias. The performance of the protonic ceramic cell was also evaluated under fuel-cell mode. With diluted CH<sub>4</sub> as the fuel, a peak power density of 0.22 W cm<sup>-2</sup> was achieved at 550 °C. The techno-economic analysis projects the LCOFP by electrochemical CO<sub>2</sub> upgrading with PCECs of 104 \$<sub>2020</sub> MWh<sup>-1</sup> at a favorable condition with a target Faradaic efficiency of 90% and an electricity cost of 0.02 \$ kWh<sup>-1</sup>. This study demonstrates a promising result for future scale-up and practical application of PCECs in electrochemical CO<sub>2</sub> upgrading as a potential grid-balancing solution.

#### CRedit authorship contribution statement

**Zehua Pan:** Conceptualization, Data curation, Formal analysis, Investigation, Methodology, Writing – original draft, Writing – review & editing, Visualization. **Chuanheng Duan:** Conceptualization, Methodology, Writing – original draft, Writing – review & editing. **Tyler Pritchard:** Methodology, Formal analysis, Investigation. **Amogh Thatte:** Methodology. **Erick White:** Funding acquisition, Writing – review & editing. **Robert Braun:** Methodology, Funding acquisition, Writing – review & editing. **Ryan O'Hayre:** Conceptualization, Supervision, Funding acquisition, Writing – review & editing. **Neal P. Sullivan:** Conceptualization, Supervision, Funding acquisition, Writing – review & editing.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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#### Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2022.121196.

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